

Effects of Boric Acid Tris-(2-oxo-[1,3]dioxolan-4-ylmethyl) Ester as an Additive on the Electrochemical Performance of Graphite Anode

Dagang Xiao¹, Jing Liu^{1*}, Rujuan Li¹, Yeping Guo², Chengxing Zong¹, Donglan Sun³ and Wei Feng^{4,5*}

¹College of Chemical Engineering and Material Science, Tianjin University of Science and Technology, Tianjin 300457, P. R. China

²Hangzhou Future Power Technology Co., Ltd, Hangzhou 311402, P. R. China

³College of Science, Tianjin University of Science and Technology, Tianjin 300457, P.R. China

⁴School of Materials Science and Engineering, Tianjin University, Tianjin 300072, P. R. China

⁵Tianjin Key Laboratory of Composite and Functional Materials, Tianjin 300072, P. R. China

Abstract

To improve the cycle performance of lithium ion battery (LIB), boric acid tris-(2-oxo-[1,3]dioxolan-4-ylmethyl) ester (BATE) as a novel electrolyte additive in 1mol·L⁻¹ LiPF₆:EC:DMC (1:1 vol., Lithium Battery-grade) electrolyte was investigated. BATE derived from glycerol carbonate, which had the same ring structure as ethylene carbonate (EC). Additionally, the carbonyl group (-C=O) and electron deficient boron atom contained in the BATE molecular structure had been proved to be effective in maintaining the stability of the lithium salt (LiPF₆) and increasing the durability of the electrolyte for formation of a solid electrolyte interface (SEI) layer, which plays a critical role in the cycle life of lithium ion battery (LIB). The electrochemical measurements and investigations of the compound BATE as additive were carried out by charge-discharge test, electrochemical impedance spectroscopy (EIS), cyclic voltammetry (CV) test and scanning electron microscope (SEM). Results confirmed that BATE decomposed on the graphite anode earlier than electrolyte solvents to form the solid electrolyte interface (SEI) film. BATE could improve cycle stability of the cell significantly, especially when addition of 1.0% BATE, the capacity retention was up to 98.1% after 20 cycles. And the internal resistance change of cell with cycles was decreased after 20 cycles.

Introduction

In the twenty-first century, electronic and information engineering has been highly developing, and portable electronic devices has been indispensable to modern life. With enhanced energy and power densities, lithium ion battery (LIB) has replaced the conventional batteries in the market for portable electronic devices. Nowadays LIB has been widely applied in electric vehicles (EVs) and hybrid electric vehicles (HEVs), especially in energy storage devices [1-4]. In order to meet the demand of human beings for high capacity and long service life, intensive efforts are still underway to further improve this battery.

As an important component of the battery, the electrolyte is one of the key factors that affect the performance and the cost of LIB. Ethylene carbonate (EC) becomes an essential electrolyte solvent since it can form an effective solid electrolyte interphase (SEI) initiated by the ring opening. And it is easy to be influenced by the surface properties of graphite [5]. Generally, carbonate-based electrolyte significantly decomposes at high voltage (>4.5 V vs. Li/Li⁺), resulting in relatively poor cycling performance for high-voltage LIB [6]. As electrolyte salt, LiPF₆ is widely used in commercial LIB. And it has been reported to react with electrolyte solvent. Solvent can induce and accelerate the decomposition of LiPF₆: LiPF₆ → LiF + PF₅, the produced PF₅ not only reacts with the solvent, but also catalyzes solvent polymerization. But some compounds can decrease the reaction activity of PF₅ by -C=O or -P=O weak binding effect [5,7-9].

One of the most economic and effective methods to overcome the above problem and to improve LIB performance is addition of electrolyte additives. The amount of an additive in the electrolyte is less than 5% either by weight or by volume; its effect on the cycleability and cycle life of LIB is significant. The electrolyte additives were discussed and summarized by researchers such as Zhang S.S. [7]. More specifically, diverse additives such as Li₂CO₃, vinylene carbonate (VC) have been reported as effective additives to improve the cycle life of LIB [10,11].

Publication History:

Received: January 13, 2016

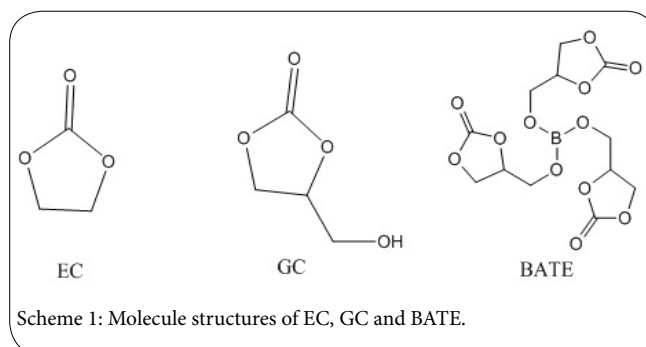
Accepted: April 20, 2016

Published: April 22, 2016

Keywords:

Boric acid tris-(2-oxo-[1,3]dioxolan-4-ylmethyl) ester, Cycleability, Electrolyte additive, Lithium ion battery, Solid electrolyte interface (SEI) film

Glycerol carbonate (GC) molecule has the same ring structure as EC shown in Scheme 1. And hydroxyl group provides a facile way to introduce various functional groups. Boron-containing compounds have been found to be effective electrolyte additives due to the lack of electrons in boron atom, and the film formed by boron-containing compounds can stabilize anion PF₆⁻ to prevent the formation of the detrimental anion F⁻ and to suppress the electrolyte further decomposition [2,12]. To combine EC structure with boron element, boric acid tris-(2-oxo-[1,3]dioxolan-4-ylmethyl) ester (BATE) was



Corresponding Author: Dr. Jing Liu, College of Chemical Engineering and Material Science, Tianjin University of Science and Technology, Tianjin 300457, P.R. China; E-mail: jingliu@tust.edu.cn

Dr. Wei Feng, College of Chemical Engineering and Material Science, Tianjin University of Science and Technology, Tianjin 300457, P.R. China; E-mail: weifeng@tju.edu.cn

Citation: Xiao D, Liu J, Li R, Guo Y, Zong C, et al. (2016) Effects of Boric Acid Tris-(2-oxo-[1,3]dioxolan-4-ylmethyl) Ester as an Additive on the Electrochemical Performance of Graphite Anode. Int J Metall Mater Eng 2: 122. doi: <http://dx.doi.org/10.15344/2455-2372/2016/122>

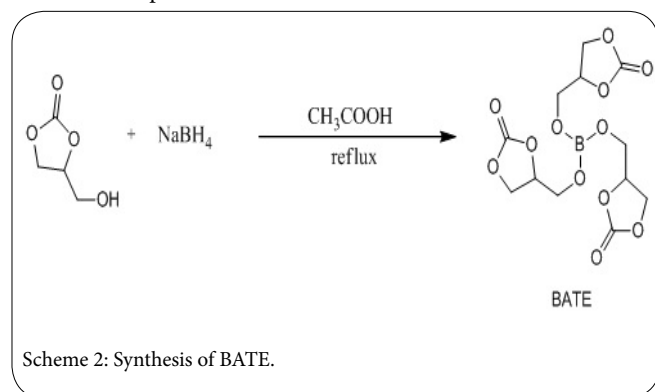
Copyright: © 2016 Xiao et al. This is an open-access article distributed under the terms of the Creative Commons Attribution License, which permits unrestricted use, distribution, and reproduction in any medium, provided the original author and source are credited.

designed for possession of their advantages (BATE chemical structure was shown in Scheme 1). And this new boron-containing compound was carefully studied as electrolyte additive in this work.

Experimental

Additive preparation

Glycerol Carbonate (GC), Sodium Borohydride (NaBH_4), Acetic acid (CH_3COOH) purchased from Jiangtian Chemical Technology Co., Ltd., Tianjin, China were analytically pure. To synthesize BATE, in a carefully dried 100 mL 3-neck round-bottom glass flask equipped with a magnetic stirrer, a reflux condenser and a pressure equalized dropping funnel, GC (17.71 g, 0.15 mol) and NaBH_4 (1.90 g, 0.05 mol) were placed. CH_3COOH (2.86 mL, 0.05 mol) was added dropwise to the slurry over a period of 15 minutes. The reaction mixture was then heated to reflux temperature for an additional 4 h until no more hydrogen was generated. After reaction, a certain amount of anhydrous dichloromethane and ethanol were added to the mixture and the precipitation was filtrated. The filtrate was evaporated under reduced pressure to give the product without further purification. BATE used in our work was synthesized by the parent compound GC in the reaction presented in Scheme 2.



Electrolyte preparation

1 mol L^{-1} LiPF_6 :EC:DMC (1:1 vol., Lithium Battery-grade) used in this work was received from Guangzhou Tinci Materials Technology Co., Ltd.. Considering the solubility and the effects of the compound on the cell electrochemical performances, electrolytes with the composition shown in Table 1 were prepared in an Ar-filled glove box (Super 1220/750/900, Mikrouna Mech. Tech. Co., Ltd., Water content: <1ppm, oxygen content: <1ppm).

Electrode preparation

Materials were obtained from Huizhou Hyperpower Batteries Inc.. Prior to use, both the Mesocarbon microbeads (MCMB) and Super-P carbon (SP) were dried at 100°C under vacuum condition for about 10 h. To prepare the electrode, a slurry consisting of uniformly mixed 90 wt.% active material (MCMB), 5 wt.% conductive additive (SP), and 5 wt.% polyvinylidene fluoride (PVDF) as a binder dispersed in 1-methyl-2-pyrrolidinone (NMP) was spread onto a $10\mu\text{m}$ thick copper foil current collector and dried overnight under a reduced pressure at 120°C . The obtained electrode sheets were compression-molded using a rolling press machine to form the active layer and punched to 13 mm in diameter. The active layer thickness was about $50\mu\text{m}$.

Electrochemical measurements

To evaluate the electrochemical properties of the electrolytes, Li/graphite half-cells (CR2025, 20 mm in diameter and 2.5 mm in thickness) were assembled in the argon-filled glove box. Each cell consisted of a MCMB working electrode, a micro porous polyolefin separator, and a lithium metal foil (0.4 mm in thickness, 15.8 mm in diameter). Charge-discharge was carried out under a constant current rate of 0.1C using a battery testing device (Neware Electronic Co. Ltd.) interfaced to a computer with software in the potential range between 0.005 V and 2.000 V (vs. Li/Li^+) at 25°C . Electrochemical impedance spectroscopy (EIS) of the cells after 2 cycles and 20 cycles were measured with an electrochemical workstation (IM6e, Zahner), data were collected in the potentiostatic mode in the 100 kHz-10 mHz frequency range at an amplitude of 5 mV. The cells were taken apart after charge-discharge test, and the MCMB electrodes were soaked in and rinsed with acetone to remove the remainder electrolyte. The morphologies of the electrodes after dried in a vacuum drier were observed by scanning electron microscope (KYKY-EM3200A, KYKY Technology Co. Ltd.). The cyclic voltammetry (CV) measurements of coin cells were conducted using a Princeton Applied Research Potentiostat/Galvanostat Model 263A electrochemical analyzer. CVs were measured within the voltage range from 0.005 V to 2.000 V (vs. Li/Li^+) at room temperature, using graphite as working electrode, Li as reference and counter electrode at a scanning rate of 0.2 mV s^{-1} .

Results and Discussion

Effect of BATE with different concentrations on the cell cyclability

The cycle performances of the graphite anode in EC-based electrolytes with different concentrations of BATE were presented in Figure 1. The graphite electrode exhibited highest cycle capacity and cycle stability for the electrolyte which contains 1.0% BATE, and the capacity retention ratio was improved by adding 2.0% BATE in the EC-based blank electrolyte, but the capacity decreased. For the blank electrolyte without BATE addition, the capacity faded rapidly. These results indicated that the addition of BATE in the EC-based electrolyte significantly enhanced the cycle performance of the graphite electrode, and BATE was feasible for use as a film-forming additive to improve the performance of commercial lithium-ion batteries. The case was similar to tris(trimethylsilyl) borate (TMSB) as an additive in EC-based electrolyte [13]. Electrolyte containing 0.5%

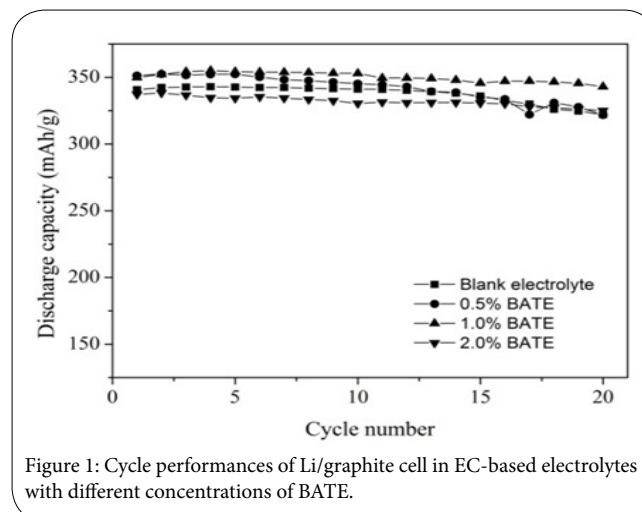


Figure 1: Cycle performances of Li/graphite cell in EC-based electrolytes with different concentrations of BATE.

BATE showed high capacity during the first several cycles, nevertheless, its capacity retention ratio was not high as expected. This may be related to SEI film did not fully form on the graphite anode. Based on the above results, the addition of BATE to electrolyte should be controlled.

Effect of BATE with different concentrations on the cell cycling efficiency

Figure 2 showed the cycling efficiency of Li/graphite cell in EC-based electrolytes with different concentrations of BATE. For organic electrolyte additives, electrochemical decomposition or co-intercalation with graphite anode frequently happened during charge-discharge cycle. If a uniform SEI layer could not form, coulombic efficiency of cycling would be low [14]. However, we found that BATE was quite compatible with graphite anode. The initial efficiency with addition of BATE was comparable to that with blank electrolyte, and the charge-discharge efficiency increased after the second cycle. As shown in Figure 2, the cycling efficiency with BATE was higher than that with blank electrolyte, and it increased as BATE concentration increased.

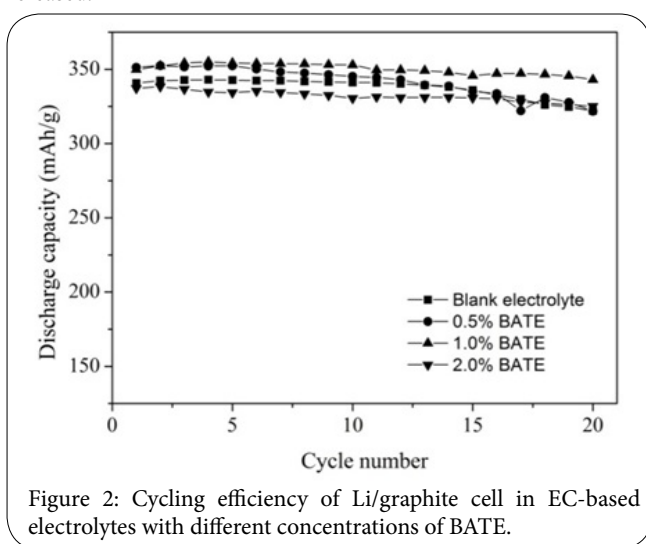


Figure 2: Cycling efficiency of Li/graphite cell in EC-based electrolytes with different concentrations of BATE.

Effect of BATE with different concentrations on the cell impedance

During cycling an internal impedance of cell increased. And that would cause capacity fading. In addition to the charge-discharge test, impedance measurements were also performed during different cycle, the Nyquist plots were presented in Figure 3. The impedance plots of Li/graphite cells with different electrolytes (Figure 3a) exhibited similar impedance behaviors: a semicircle at high frequencies reflected the interfacial impedance of SEI and charge-transfer resistances, and a slop line at low frequencies was assigned as the Warburg impedance which was associated with the ions diffusion process in the electrode materials. As the charge-transfer resistance is related to the semicircle diameter, the smaller the semicircle diameter, the smaller the charge transfer resistance [15,16]. Impedance of cell with 2.0% BATE was much higher than that with other concentrations, maybe due to high viscosity of electrolyte with too much BATE or the formation of a too thick and compact SEI film, which increased ion transfer resistance and limited the capacity utilization. Balakrishnan et al. [17] supposed that the deterioration of cell performance is owing to either electrochemical instability or additive viscosity increase (which leads to capacity fading or affects capacity utilization). The impedance plots of Li/graphite cells after 20 cycles were shown in Figure 3(b), the impedance of all cells increased with cycling. Whereas, the increase of cell with blank electrolyte and with 0.5% BATE changed significantly. This could be related to electrochemical instability for lack of a uniform SEI film formed during the first charge-discharge process.

Effect of BATE with different concentrations on the cell cyclic voltammetry

As an SEI-forming additive, the preferable reduction to electrolyte solvents is important [18]. The cyclic voltammetry curves of the Li/graphite cells with different electrolytes were displayed in Figure 4. From the partial enlarged view (inset a), obvious irreversible peaks at 1.6 V-1.9 V (vs. Li/Li⁺) were observed only in the electrolytes containing BATE. These peaks were related to the BATE reduction. From the partial enlarged view (inset b), a strong reductive irreversible peak at 0.5 V-0.8 V (vs. Li/Li⁺) in BATE-free electrolyte was observed obviously. The reductive peak was attributed to the decomposition of EC molecules and associated with the formation of SEI film on the

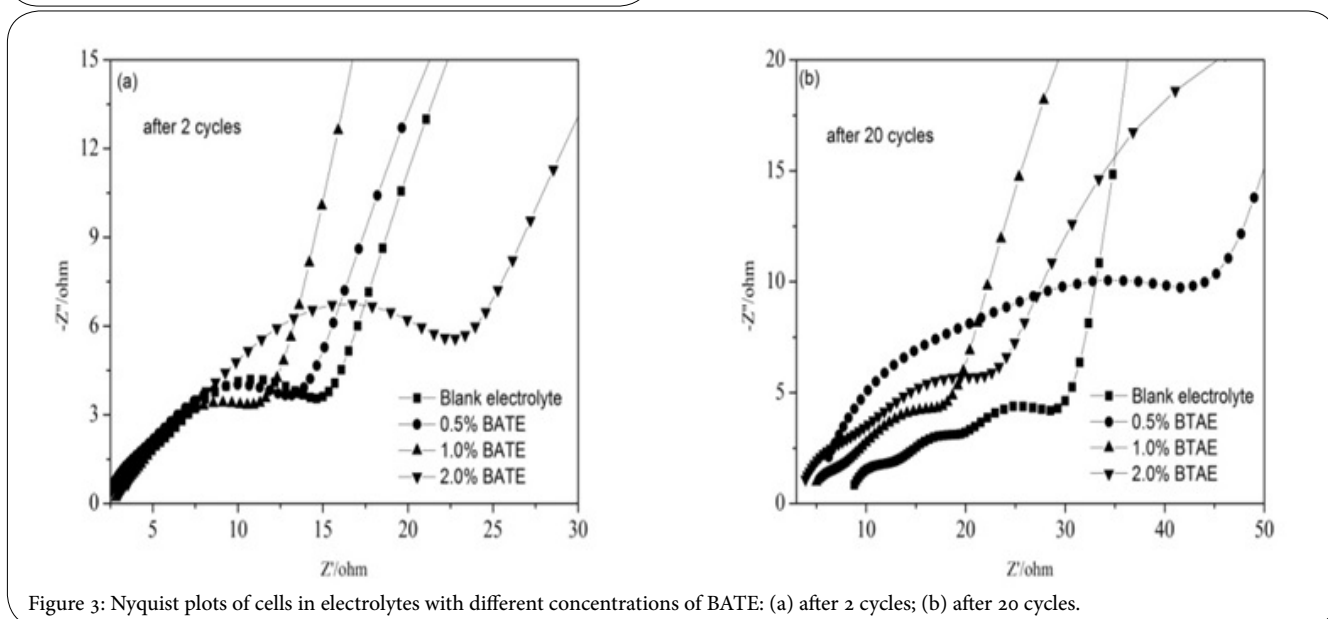


Figure 3: Nyquist plots of cells in electrolytes with different concentrations of BATE: (a) after 2 cycles; (b) after 20 cycles.

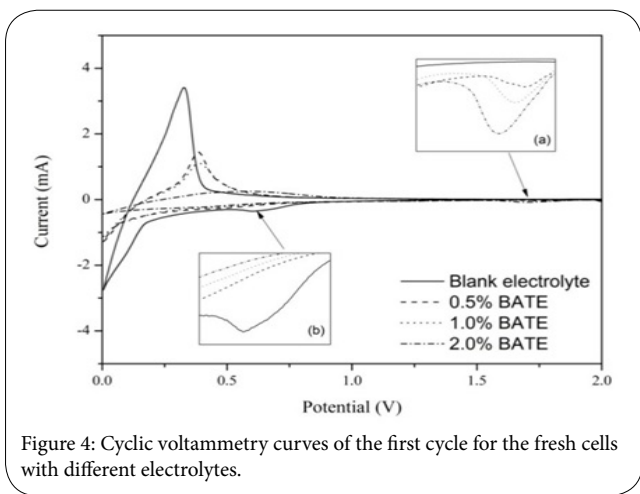


Figure 4: Cyclic voltammety curves of the first cycle for the fresh cells with different electrolytes.

electrolyte [19]. The cyclic voltammety curves shown in Figure 5 present the change of reversible peak current density before and after charge-discharge test. Compared with the curves of cells with different electrolytes in Figure 5(a) and 5(b), current density of the reversible peak for blank electrolyte showed a great decrease, so it is the same case for 0.5% BATE electrolyte. However, current density of the reversible peak were relatively stable in electrolytes with 1.0% and 2.0% BATE respectively, which also indicated the good cycleability in the charge-discharge test.

Effect of BATE with different concentrations on graphite anode morphology

Figure 6 presented SEM micrographs of the graphite electrode surface before and after cycled in different electrolytes. Before cycling, the graphite particles with a little adhesive/conductive agent could be identified in the pristine electrode as shown in Figure 6(a). After 20

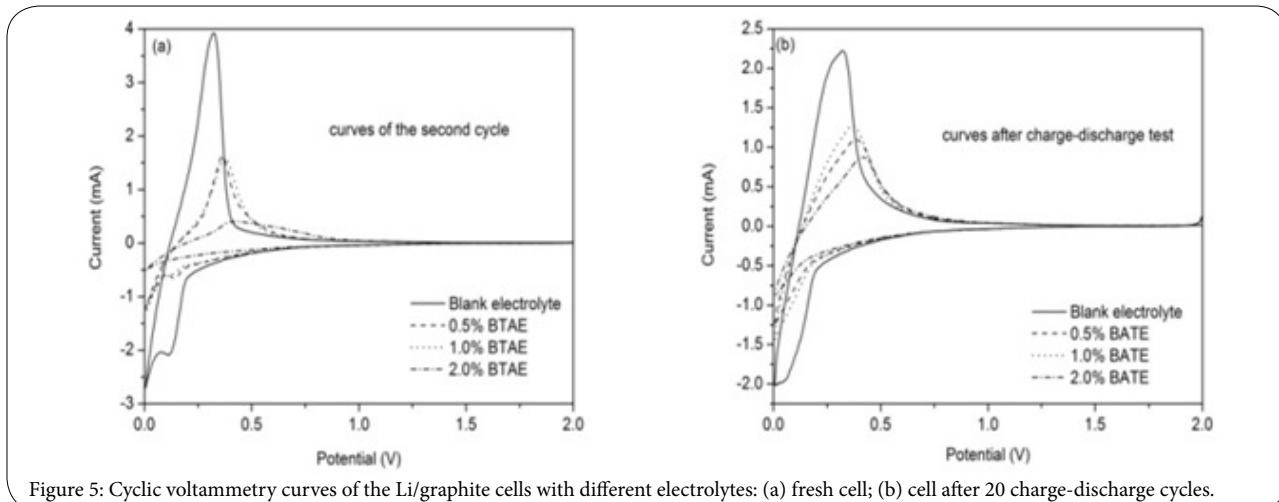


Figure 5: Cyclic voltammety curves of the Li/graphite cells with different electrolytes: (a) fresh cell; (b) cell after 20 charge-discharge cycles.

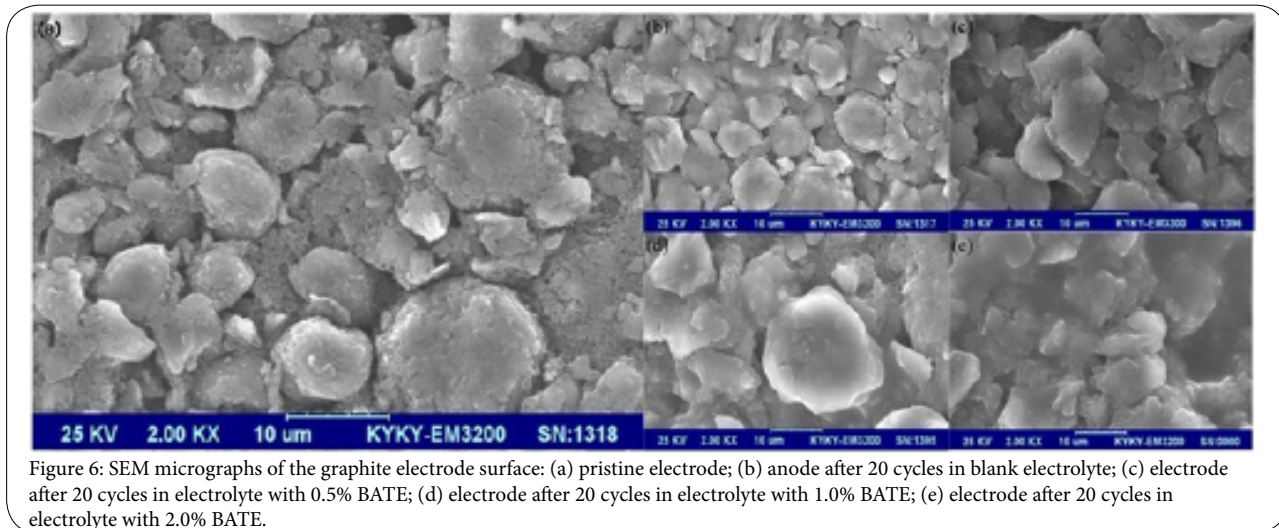


Figure 6: SEM micrographs of the graphite electrode surface: (a) pristine electrode; (b) anode after 20 cycles in blank electrolyte; (c) electrode after 20 cycles in electrolyte with 0.5% BATE; (d) electrode after 20 cycles in electrolyte with 1.0% BATE; (e) electrode after 20 cycles in electrolyte with 2.0% BATE.

graphite electrode surface. Peak of cell in the electrolyte with BATE disappeared at 0.5 V-0.8 V (vs. Li/Li⁺), which meant that BATE could mitigate EC decomposition at approximately 0.6 V and increase the electrolyte durability.

Current density of the reversible peak barely decreased during cycle, which implied the good cycleability of graphite electrode in the

cycles, the anode in blank electrolyte without BATE was covered by a film. So the graphite particles could not be observed any more, of which the cell exhibited a faster capacity fading in the charge-discharge test, as shown in Figure 6(b). At the graphite anode, surface fractures in graphite particles can be induced with cycles, the subsequent surface area increase and electrolyte reduction reactions that consume more cycleable Li [20, 21]. The morphology of the graphite electrode after 20 cycles with 0.5% BATE electrolyte was

similar to the blank electrolyte (Figure 6(c)), which acted the same behavior in the cycling test as cell with blank electrolyte, owing to the incomplete SEI film formed by small BATE. Morphologies of electrodes with 1.0% and 2.0% BATE electrolyte were uniform, and the surfaces of the electrodes were covered with compact film (Figure 6(d) and (e)), which indicated that a well stable SEI was formed [22]. It seemed that a thick SEI film could prevent the graphite anode from direct contacting with the electrolyte more adequately. However, a too thick and compact film can deteriorate the cell performance because it may retard Li^+ transfer and decrease the capacity performance [23]. So amount of addition of additive to electrolyte should be controlled.

Conclusion

In this study, the compound of BATE, a glycerol carbonate derivative, added in 1 mol L^{-1} LiPF_6 :EC:DMC (1:1, v:v) as electrolyte additive was investigated by charge-discharge test, electrochemical impedance test, cyclic voltammetry test and scanning electron microscope test. As an electrolyte additive, BATE obviously improved cycle performance of the cell. Especially cell with 1.0% BATE electrolyte had highest capacity and highest capacity retention in long cycles, and it had higher cycling efficiency than cell with blank electrolyte. The improvements attributed to a stable and protective SEI film with a low resistance on the anode surface formed by BATE reduction prior to electrolyte solvents. The SEI film could suppress electrolyte decomposition and prevent graphite electrode from destruction effectively.

Competing Interests

The authors have declared that no competing interests exist.

Author Contributions

All the authors substantially contributed to the study conception and design as well as the acquisition and interpretation of the data and drafting the manuscript.

Funding

This work was financially supported by the National Natural Science Foundation of China (Grant No.21373150).

References

1. Verma P, Maire P, Novák P (2010) A review of the features and analyses of the solid electrolyte interphase in Li-ion batteries. *J Electrochimica Acta* 55: 6332-6341.
2. Wang XS, Xing LD, Liao XL, Chen XF, Huang WN, et al. (2015) Improving cyclic stability of lithium cobalt oxide based lithium ion battery at high voltage by using trimethylboroxine as an electrolyte additive. *J Electrochimica Acta* 173: 804-811.
3. Li Y, Lian F, Ma LL, Liu CL, Yang L, et al. (2015) Fluoroethylene Carbonate as Electrolyte Additive for Improving the electrochemical performances of High-Capacity $\text{Li}_{1.16}[\text{Mn}_{0.75}\text{Ni}_{0.25}\text{O}_2]$ Material. *J Electrochimica Acta* 168: 261-270.
4. Yang JP, Zhao P, Shang YM, Wang L, He XM, et al. (2014) Improvement in High-voltage Performance of Lithium-ion Batteries Using Bismaleimide as an Electrolyte Additive. *J Electrochimica Acta* 121: 264-269.
5. Zhang SS, Xu K, Jow TR (2006) EIS study on the formation of solid electrolyte interface in Li-ion battery. *J Electrochimica Acta* 51: 1636-1640.
6. Xu MQ, Liu YL, Li B, Li WS, Li XP, et al. (2012) Tris (pentafluorophenyl) phosphine: An electrolyte additive for high voltage Li-ion batteries. *J Electrochemistry Communications* 18:123-126.
7. Zhang SS (2006) A review on electrolyte additives for lithium-ion batteries. *J Power Sources* 162:1379-1394.
8. Victor A. Agubra, Jeffrey W. Fergus (2014) The formation and stability of the solid electrolyte interface on the graphite anode. *J Power Sources* 268:153-162.
9. Zhang SS, Xu K, Jow TR (2003) Tris(2,2,2-trifluoroethyl) phosphite as co-solvent for nonflammable electrolytes in Li-ion battery. *J Power Sources* 113:166-172.
10. Shin JS, Han CH, Jung UH, Lee SI, Kim HJ, et al. (2002) Effect of Li_2CO_3 additive on gas generation in lithium-ion batteries. *J Power Sources* 109:47-52.
11. Aurbach D, Gamolsky K, Markovsky B, Gofer Y, Schmidt M, et al. (2002) On the use of vinylene carbonate (VC) as an additive to electrolyte solutions for Li-ion batteries. *J Electrochimica Acta* 47: 1423-1439.
12. Zhu Y, Li Y, Bettge M, P. Abraham D (2013) Electrolyte additive combinations that enhance performance of high-capacity $\text{Li}_{1.2}\text{Ni}_{0.15}\text{Mn}_{0.55}\text{Co}_{0.1}\text{O}_2$ -graphite cells. *J Electrochimica Acta* 110: 191-199.
13. Liu YB, Tan L, Li L (2013) Tris(trimethylsilyl) borate as an electrolyte additive to improve the cyclability of LiMn_2O_4 cathode for lithium-ion battery. *J Power Sources* 221: 90-96.
14. Feng JK, Cao YL, Ai XP, Yang HX (2008) Tri-(4-methoxyphenyl) phosphate: A new electrolyte additive with both fire-retardancy and overcharge protection for Li-ion batteries. *J Electrochimica Acta* 53: 8265-8268.
15. Sultana I, Mokhesur RM, Wang JZ, Wang CY, Wallace GG, et al. (2012) All-polymer battery system based on polypyrrole (PPy)/para (toluene sulfonicacid) (pTS) and polypyrrole (PPy)/indigo carmine (IC) free standing films. *J Electrochimica Acta* 83: 209-215.
16. Shao YY, Engelhard M, Lin YH (2009) Electrochemical investigation of polyhalide ion oxidation-reduction on carbon nanotube electrodes for redox flow batteries. *J Electrochemistry Communications* 11:2064-2067.
17. Balakrishnan PG, Ramesh R, Kumar TP (2006) Safety mechanisms in lithium-ion batteries. *J Power Sources* 155: 401-414.
18. Li B, Wang YQ, Lin HB, Wang XS, Xu MQ, et al. (2014) Performance improvement of phenyl acetate as propylene carbonate-based electrolyte additive for lithium ion battery by fluorine-substituting. *J Power Sources* 267:182-187.
19. Wang HY, Masaki Yoshio (2001) Carbon-coated natural graphite prepared by thermal vapor decomposition process, a candidate anode material for lithium-ion battery. *J Power Sources* 93:123-129.
20. Song HS, Cao Z, Zhang ZA, Lai YQ, Li J, et al. (2014) Effect of vinylene carbonate as electrolyte additive on cycling performance of LiFePO_4 /graphite cell at elevated temperature. *J Transactions of Nonferrous Metals Society of China* 24: 723-728.
21. Markervich E, Salitra G, Levi MD, Aurbach D (2005) Capacity fading of lithiated graphite electrodes studied by a combination of electroanalytical methods, Raman spectroscopy and SEM. *J Power Sources* 146:146-150.
22. An YX, Zuo PJ, Cheng XQ, Liao LX, Yin GP (2011) The effects of LiBOB additive for stable SEI formation of PP13TFSI-organic mixed electrolyte in lithium ion batteries. *J Electrochimica Acta* 56:4841-4848.
23. Abe K, Yoshitake H, Kitakura T, Hattori T, Wang HY, et al. (2004) Additives-containing functional electrolytes for suppressing electrolyte decomposition in lithium-ion batteries. *J Electrochimica Acta* 49:4613-4622.